

Long-Range Refractive Index Sensing Using Plasmonic Nanostructures

Tomas Rindzevicius, Yury Alaverdyan, and Mikael Käll*

Department of Applied Physics, Chalmers University of Technology, SE-412 96 Göteborg, Sweden

W. Andrew Murray and William L. Barnes

School of Physics, University of Exeter, Exeter, EX4 4QL United Kingdom

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We study the variation in localized surface plasmon resonance (LSPR) spectra ($\lambda_{\text{LSPR}} \approx 600$ nm) as a function of dielectric coating thickness. Langmuir–Blodgett multilayers composed of 22-tricosenoic acid ($n \approx 1.53$) were deposited onto short-range-ordered Au nanodisks and nanoholes on glass in air. For large dielectric coating thicknesses ($d = 100$ – 340 nm), the LSPRs exhibit a pronounced oscillatory behavior with a periodicity of ~ 190 nm. This is in agreement with a simple image-dipole model, which yields a periodicity of $\sim \lambda_{\text{LSPR}}/2n$. However, the amplitude of the dipolar plasmon resonance wavelength oscillation $\lambda_{\text{LSPR}}(d)$ is surprisingly large, of the order 25–45 nm for $d \approx 300$ nm, indicating the importance of finite size effects. The large LSPR shifts observed at such a large distance from the actual metal surface suggest the possibility of using comparatively thick dielectric films as spacer layers in bio/chemo LSPR sensor applications.

1. Introduction

It is well-known that excitation of localized surface plasmon resonances (LSPRs) in noble metal particles or nanostructures result in intense extinction bands that are absent in the bulk or for flat surfaces.¹ The LSPR wavelength, intensity, and line-width depends on the size, shape, and dielectric properties of the nanostructure, as well as on its dielectric environment and the presence of nearby nanostructures.^{1–3} This dependence allows for fine-tuning of various nanoparticle architectures for use as functional materials in numerous applications, including, but not limited to, surface enhanced spectroscopies,^{4–7} cancer therapy,⁸ and chemo/bio sensors.^{9–16} The latter application is of special interest, because it may allow for label-free detection of extremely small concentrations of target molecules. Recent nanoparticle-based sensing mechanisms include nanoparticle aggregation,^{9,15,17,18} surface enhanced Raman scattering,¹⁶ and refractive index variation detection.^{11–14} The latter sensing principle, which relies on adsorbate induced changes in the local refractive index at the noble metal surface, is the most easily realized in practice. The strong confinement of the electromagnetic fields allows one to miniaturize the LSPR-based refractive index sensors down to the single particle level,¹² where spectral shifts corresponding to zeptomol concentration changes can be detected. The absolute sensitivity to bulk refractive index (RI) changes differs to a great extent between different particle classes (e.g., between spherical particles, nanodisks, nanorods, and nanoshells), but much of this variation seems to be due to a linear scaling between sensitivity and LSPR wavelength.¹⁹ In the case of dielectric-coated metallic particles, it is usually assumed that the LSPR shift exhibits an essentially exponential decay until saturation sets in for dielectric layer thicknesses comparable to the particle size.^{14,20} The magnitude of the LSPR shift is then determined by the thickness of the dielectric layer, d , the contrast between the refractive indices of the layer and

the bulk environment,²⁰ and the decay length of the E-field induced through LSPR excitation. This short-distance dependence of the LSPR has been experimentally quantified for single nanoparticles,¹² weakly interacting triangular silver particles prepared by nanosphere lithography (NSL),^{13,14} and nanoholes in optically thin gold films.²¹ Using self-assembled monolayers (SAMs), the LSPR shift of the Ag nanoparticles was found to saturate at $d \approx 40$ nm. These kinds of experiments are clearly important in order to tune the sensing capabilities of nanoparticles to match the dimensions of biological analytes.

However, it is well-known that the LSPR spectrum not only depends on the dielectric environment in the immediate vicinity of the metal surface. If the surrounding medium is such that the LSPR induced far-field is coherently fed back to the plasmonic nanostructure, a substantial LSPR shift and line-shape change can occur even at large distances. Such diffractive far-field coupling has been observed for ordered arrays and lines of nanoparticles^{22,23} and for particles placed above a reflecting metal surface.²⁴ In a recent paper, Murray et al. showed that the latter effect can also be observed if Ag nanoparticles fabricated by NSL are placed near an interface between two dielectric media.²⁵ It was observed that a substantial LSPR shift (amplitude ~ 20 nm) remained at distances up to $d \approx 300$ nm from the metal surface. This effect is analogous to the lifetime oscillations observed for fluorescent molecules near an interface between two media,²⁶ which can be interpreted in terms of so-called cavity quantum electrodynamics. The observation by Murray et al. clearly motivates further investigations of the long-range distance dependence of the LSPR in metal nanostructures. In this paper, we study such effects in short range ordered Au nanodisks and nanoholes prepared by colloidal lithography. Both types of structures are highly interesting for bio/chemo sensing applications.^{21,27–29} We show that the LSPR position and line-shape are dramatically affected when the nanostructures are coated with thick (100–340 nm) dielectric layers. The experimental results are discussed in terms of the classical image-dipole model, which is in good qualitative agreement with data.

* To whom correspondence should be addressed. E-mail: kall@fy.chalmers.se.

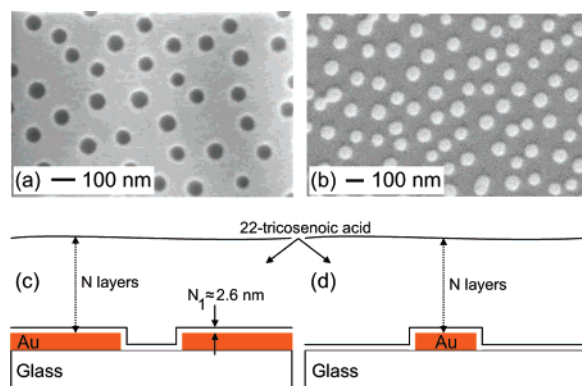


Figure 1. Representative SEM images of ~ 60 nm short range ordered arrays of nanoholes (density $\sim 40 \mu\text{m}^{-2}$) in a 20 nm Au film on glass (a) and ~ 70 nm Au nanodisks (density $\sim 50 \mu\text{m}^{-2}$) on glass (b) together with schematics of the nanostructures coated with N layers of 22-tricosenoic acid (c).

87 Further, the rapid change in the LSPR spectrum for large layer
 88 thicknesses indicates that it is possible to detect target molecules
 89 situated hundreds of nanometers away from the metal particle
 90 surface. This observation can be of high practical importance
 91 for the development of novel nanoplasmonic chemo/bio sensors.

92 2. Experimental Section

93 The short range ordered arrays of nanometric holes and disks
 94 on glass substrates were fabricated using colloidal lithography,²⁷
 95 which is capable of producing large areas ($> \text{cm}^2$) of nano-
 96 structures. In the case of nanoholes, colloidal polystyrene spheres
 97 (sulfate latex, IDC U.S.A.) were adsorbed on a glass substrate,
 98 and a 20 nm gold film was evaporated on the surface. The Au
 99 covered colloidal polystyrene spheres were then removed by
 100 tape stripping. The result is a 20 nm thick Au film containing
 101 circular apertures with glass at the bottom; see illustrative SEM
 102 image in Figure 1a. In preparing nanometer Au disks, the
 103 colloidal polystyrene spheres were instead adsorbed after the
 104 20 nm gold film has been deposited.³⁰ A heat treatment is then
 105 applied to flatten out the polystyrene spheres. The gold metal
 106 that is not covered by polystyrene is removed using argon ion
 107 beam etching. Finally, the polystyrene particles are removed
 108 by tape stripping. This results in 20 nm high circular Au disks,
 109 see Figure 1b. The nanohole and nanodisk diameters are
 110 controlled by the polystyrene sphere size, and the distances
 111 between the objects are tuned by choosing different sphere
 112 densities. Due to the heat treatment used for fabricating Au
 113 nanodisks, the disk diameters were larger than the diameters of
 114 the nanoholes, i.e., ~ 70 nm versus ~ 60 nm. The nanohole and
 115 nanodisk densities were ~ 40 and $\sim 50 \mu\text{m}^{-2}$, respectively. This
 116 choice of parameters results in very similar LSPR wavelengths,
 117 i.e., $\lambda_{\text{LSPR}} \approx 600$ nm, for the disks and the holes. However,
 118 it is not possible to also obtain the same line-width for the two
 119 types of samples due to intrinsic differences in LSPR decay
 120 processes. Specifically, the line-width of the nanodisk resonance
 121 is determined by a combination of radiation damping and
 122 internal dissipation,³⁰ while the hole resonance is also affected
 123 by coupling to extended surface plasmon polariton modes of
 124 the thin film.^{28,31}

125 The Langmuir–Blodgett (LB) technique was used to sequen-
 126 tially deposit multilayer stacks of 22-tricosenoic acid on the
 127 plasmonic nanostructures.³² A layer of 22-tricosenoic acid from
 128 a chloroform solution was first spread on a water surface. The
 129 layer was then slowly compressed to a surface pressure of about
 130 30 mNm^{-1} , which results in monolayer formation. The nanohole

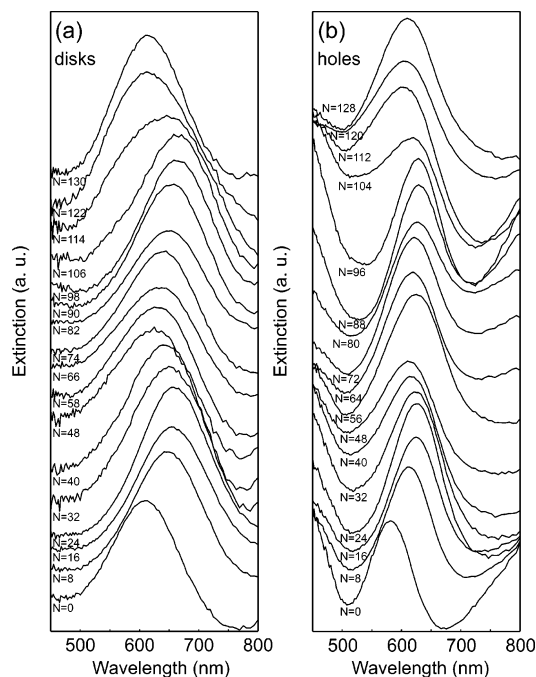


Figure 2. Extinction spectra of gold nanodisks (a) and nanoholes (b) as the number of dielectric layers is sequentially increased from $N = 0$ to 130 ($d = 0$ to ~ 350 nm). The thickness of one 22-tricosenoic acid layer is approximately 2.6 nm.

or nanodisk samples were positioned normal to the water surface
 and then dipped vertically downward and upward at a rate of
 0.25 mm/s while the surface pressure was adjusted to remain
 constant throughout the whole sample dipping procedure. The
 number of deposited layers varied from 0 to 140. The latter
 corresponds to a thickness of ~ 364 nm. LB films deposited in
 this way on nanohole films have been studied before.³³ In that
 study, it was found that the LB film tends to bridge the
 nanoholes. Further, in that study, which looked at fluorescence
 from LB film over-coated nanohole films, the assumption of a
 planar, homogeneous LB film was found to fit the data well;
 we have therefore assumed the LB film deposits on our samples
 in the same way.

In the case of the nanohole structures, the sample contained
 36 regions that contained a different number of layers and one
 uncoated reference region. The reproducibility was checked by
 coating three identical samples with 0–6 layers and then
 comparing the results. No significant differences were found.
 In the case of the short range ordered Au nanodisk structures,
 nine identical samples were used. Each sample contained 5
 regions with increasing overlayer coating thickness, and one
 uncoated reference region to allow for determination of the
 relative LSPR shift.

The optical properties of the hole and disk samples were
 quantified using normal incidence extinction spectroscopy. A
 collimated beam with a divergence of $\sim 0.5^\circ$ and a diameter of
 ~ 1 mm from a tungsten halogen lamp was directed onto the
 nanostructures. The transmitted intensity was measured vs
 wavelength using a phase-sensitive lock-in technique. For a
 more detailed description, see ref 25. The beam diameter
 corresponds to averaging over $\sim 1.4 \times 10^4$ scattering objects.

3. Results and Discussion

3.1. Experimental Results. Extinction spectra for the Au
 nanodisk samples as a function of the sequentially increasing
 number of dielectric layers N are presented in Figure 2a. As
 expected, the LSPR spectra first exhibit a rapid red-shift with

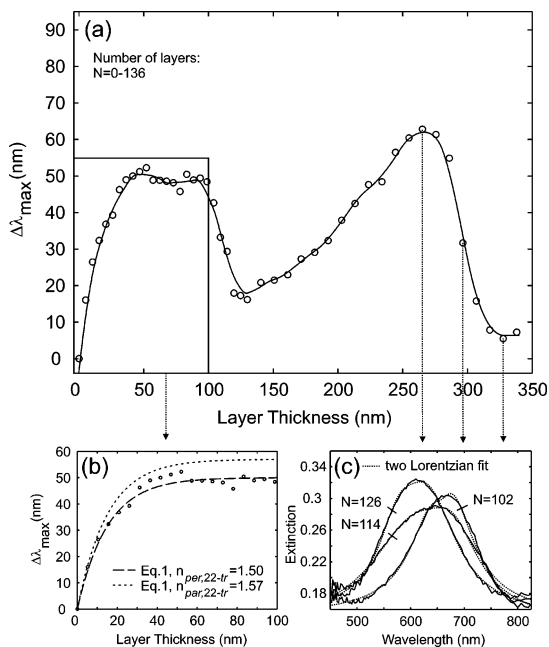


Figure 3. Summary of LSPR variation induced by an increasing layer thickness for gold nanodisks. (a) Relative LSPR shift plotted as a function of the dielectric layer thickness d . The solid line is a guide for the eye only. (b) Comparison between experiment and the exponential decay modes for the LSPR shift, using two refractive index values for the dielectric layer, in the region $d = 0-100$ nm. (c) Illustration of line-shape changes and broadening effects in the region of rapid blue-shifting.

167 increasing coating thickness. However, for $N \geq 40$ and $d \geq$
 168 100 nm, where d is the total coating thickness, the LSPR spectra
 169 begin to blue-shift and then exhibit a periodic oscillation. This
 170 behavior is accompanied by a fluctuating peak shape. In
 171 particular, the line width of the LSPR becomes anomalously
 172 broad when the blue-shifting occurs. The same kinds of effects
 173 are seen for the short range ordered nanohole structures, see
 174 Figure 2b. Before analyzing the optical response for thick
 175 dielectric layers ($d > 100$ nm), it is important to focus on the
 176 well-studied region ($d < 100$ nm) and compare the results to
 177 previous experimental studies.³⁴ In Figure 3a, the LSPR shift
 178 in relation to the uncoated case, $\Delta\lambda_{\max}(d)$, for the nanodisk
 179 samples, is shown. In many cases,³⁵⁻³⁸ the short range sensitivity
 180 to refractive index changes can be approximated by a model
 181 for surface plasmons on a planar metal surface. The LSPR
 182 wavelength shift is then expected to decrease exponentially as
 183 a function of coating thickness according to

$$\Delta\lambda_{\max} = m\Delta n[1 - \exp(-d/L)] \quad (1)$$

184 Here m is the refractive index sensitivity $\delta\lambda_{\max}/\delta n$, obtained
 185 from measurements in different uniform refractive index
 186 environments, Δn is the difference in refractive indices between
 187 the thin layer and the surrounding medium, d is the layer
 188 thickness, and L is the characteristic decay length of the local
 189 electromagnetic field surrounding the plasmonic nanostructure.
 190 Using eq 1, the Au nanodisk LSPR shift can be well reproduced
 191 using previously measured values for the different parameters,
 192 i.e., the thickness of the adsorbate layer being $d = N2.6$ nm,³⁹
 193 $m = 100$ nm,³⁰ $\Delta n = 0.50$, or 0.57 and $L = 15$ nm;^{34,40} see
 194 Figure 3b. The two values for Δn are motivated by the fact
 195 that 22-tricoscenoic acid is anisotropic, with $n = 1.57$ parallel
 196 and $n = 1.50$ perpendicular to the substrate, respectively. Both
 197 values give good agreement with the experimental data, as can
 198 be seen in Figure 3b. Equation 1 predicts that the LSPR shift

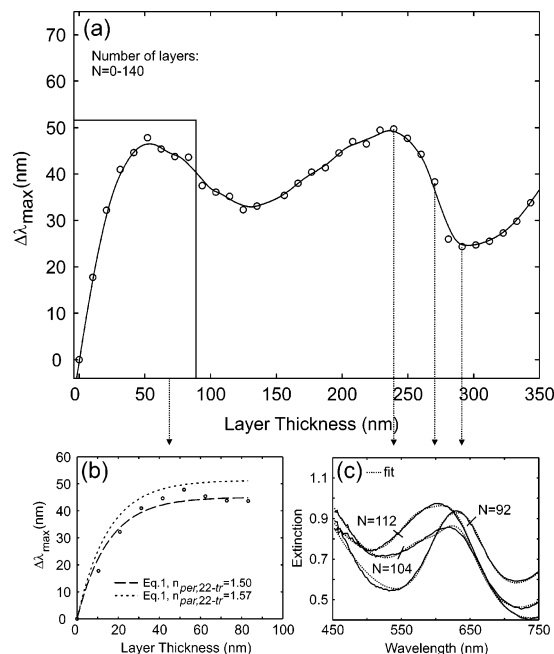


Figure 4. Same as in Figure 3 but for the short range ordered nanoholes in a 20 nm gold film.

should saturate at $\Delta\lambda_{\max} = m\Delta n \approx 50-57$ nm for $d \geq 50$ nm. 199
 However, this is clearly not the case. For $d > 100$ nm, the LSPR 200
 displays a dramatic oscillatory behavior, with a periodicity and 201
 amplitude of $d_p \approx 190$ nm and $\Delta\lambda_{\text{amp,max}} \approx 45$ nm, respectively. 202
 It is also clear that the oscillation is anharmonic; that is, the 203
 plasmon shift is much faster in the blue-shifting regions (at 204
 around $d \approx 120$ and 280 nm) than in the region between ~ 150 205
 and 250 nm, see Figure 3a. A closer look at the spectra in Figure 206
 2 shows that these fast shifting regions are also characterized 207
 by broader and asymmetric line shapes. This is highlighted in 208
 Figure 3c, which compares spectra for $N = 102, 114,$ and $126.$ 209

Figure 4a shows the variation in the nanohole extinction peak 210
 maximum obtained from the spectra displayed in Figure 2b. 211
 As in the case of the gold nanodisks, the $d < 100$ nm region 212
 can be well described using eq 1, see Figure 4b. A refractive 213
 index sensitivity $m = 90$ nm and a characteristic decay length 214
 $L = 15$ nm from ref 21 obviously yield good agreement with 215
 experimental data. For $d > 100$ nm, $\Delta\lambda_{\max}$ again exhibits an 216
 oscillatory behavior with a period close to $d_p \approx 180$ nm, similar 217
 to the result for the gold nanodisks. Although the amplitude of 218
 the LSPR oscillations for the nanohole structure $\Delta\lambda_{\text{amp,max}} \approx$ 219
 25 nm is somewhat smaller than for disks, one notes a similar 220
 anharmonic variation in $\Delta\lambda_{\max}(d)$. One also notes a similar 221
 broadening in the blue-shifting regions. 222

Figure 5a,b shows the numerical derivatives of $\Delta\lambda_{\max}$ with 223
 respect to the layer thickness d . The most sensitive thickness 224
 regions, corresponding to the rapid blue-shift and marked with 225
 s in the graph, occur at $d \approx 100$ nm and $d \approx 290$ nm for the 226
 disks and at slightly lower thicknesses for the holes. We note 227
 that the sensitivities in these fast-shifting regions, of the order 228
 $\delta\Delta\lambda_{\max}/\delta d \approx 1$ or above, are only a factor of 2-3 lower than 229
 what is found close to the metal surface.²¹ This may have 230
 implications for novel LSPR sensor constructs, as mentioned 231
 below. 232

3.2. Interpretation. The LSPR $\Delta\lambda_{\max}(d)$ curves for the two 233
 types of nanostructures clearly display very similar patterns. In 234
 the following, we will focus on the optical response of the gold 235
 nanodisks, for which the assignment of the optical resonance 236
 to a dipolar LSPR is well established.¹ This allows one to 237

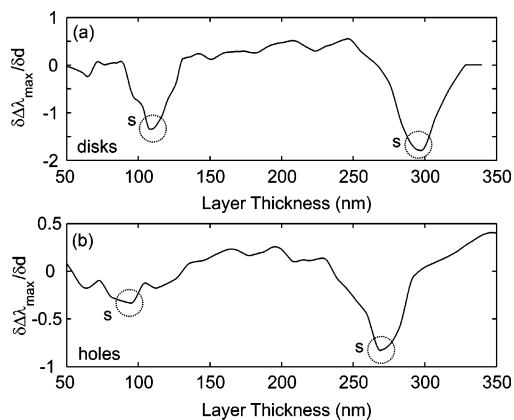


Figure 5. Derivative of the experimental $\Delta\lambda_{\max}(d)$ values with respect to d for (a) nanodisks and (b) nanoholes. The part of the graph marked with the sign s indicates regions that are the most sensitive to a change in the dielectric layer thickness.

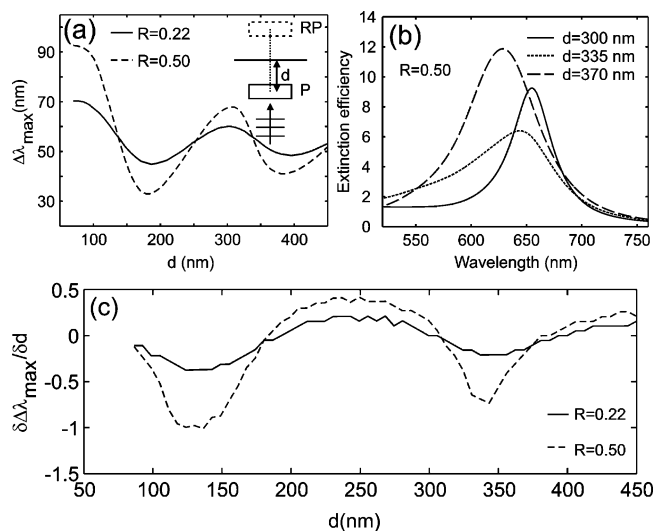


Figure 6. (a) Theoretical prediction of the extinction peak shift $\Delta\lambda_{\max}$ as a function of dielectric layer thickness, d , from the image-dipole model, which is shown schematically in the inset. (b) Illustration of extinction spectra for $R = 0.50$ and d values within the blue-shifting region. (c) Theoretical derivatives of the $\Delta\lambda_{\max}$ versus d curves is plotted versus layer thickness for $R = 0.22$ and 0.50 .

238 formulate a simple physical picture of the observed phenomena
 239 using the image-dipole model, illustrated in the inset of Figure
 240 6c. A similar approach was used by Holland et al.²⁴ to interpret
 241 LSPR frequency shifts observed for Ag and Au islands placed
 242 near a silver surface. We assume that the refractive indices of
 243 the LB film ($n = 1.50\text{--}1.57$) and the glass substrate ($n = 1.52$)
 244 are so close that we can neglect multiple reflections between
 245 the LB–air and the LB–glass interfaces. We then approximate
 246 the nanodisk as a point dipole oscillator that is affected by the
 247 field reflected back from the dielectric–air interface. The dipole
 248 moments for the particle P_P and its image P_I can then be defined
 249 as

$$P_P = \alpha(E_0(1 + R \cos(k2d)) + AP_I)$$

$$P_I = RP_P \quad (2)$$

250 Here, α is the nanodisk polarizability, E_0 is the incident field,
 251 $AP_I = (1/(2d)^3 - ik/(2d)^2 - k^2/2d) \exp(ik2d)P_I$ is the image
 252 dipole field, which simulates the field reflected back to the
 253 nanodisk due to the 22-tricosenoic acid–air interface, $R = 0.22$
 254 is the amplitude reflection coefficient at the dielectric–air

interface, d is the distance to the interface, and $k = \omega/c_0n$ is the
 wavevector of the optical field within the dielectric medium.
 The coupled dipole equation leads to an expression for the
 effective particle polarizability

$$\alpha_{\text{eff}} = \alpha \frac{1 + R \cos(k2d)}{1 - \alpha AR} \quad (3)$$

A gold nanodisk is then approximated by an oblate spheroid
 and the polarizability is corrected for radiation damping.³⁰ This
 yields

$$\alpha = \frac{\tilde{\alpha}}{1 - \frac{ik^3\tilde{\alpha}}{6\pi}} \quad (4)$$

where $\tilde{\alpha}$ is the quasi-static spheroid polarizability given by¹

$$\tilde{\alpha} = 4\pi a^2 c \frac{\epsilon_s - \epsilon_m}{3\epsilon_m + 3L(\epsilon_s - \epsilon_m)} \quad (5)$$

Here, a and c are the long and the short axis radii of the spheroid,
 ϵ_s is the complex dielectric function of gold, and $\epsilon_m = n^2$ is the
 dielectric constant of the surrounding medium. L is the spheroid
 shape factor, which can be calculated from the disk aspect ratio
 a/c .¹ In order to compare the theoretical model with the
 experimental extinction spectra, we calculate the effective
 extinction efficiency for various layer thicknesses d . This is
 given by the sum of the extinction efficiencies for the particle
 and its image, which yields

$$C_{\text{ext}} = k(1 + R)\text{Im}[\alpha_{\text{eff}}]/(\pi a^2) \quad (6)$$

Figure 6 shows the LSPR shift vs d for an oblate Au spheroid
 that is 70 nm in diameter and 20 nm in height using the
 formalism described above. The dielectric function for gold was
 taken from Johnson and Christy,⁴¹ and the refractive index of
 the dielectric environment was set to $n = 1.52$. The reference
 ($d = 0$) LSPR position, which defines $\Delta\lambda_{\max} = 0$, is obtained
 from eq 4 using the average refractive index of glass and air,
 as the image-dipole model is not valid for very small d values.
 As is shown in Figure 6a (full line), the calculated $\Delta\lambda_{\max}$ curve
 oscillates with a periodicity of ~ 200 nm, in good agreement
 with the experiment. However, the amplitude of the oscillation
 is significantly smaller than the measured one. As shown by
 the dashed curve in Figure 6a, a much better agreement is found
 if the reflection coefficient R is increased from $R = 0.22$ to
 0.50. This change also results in an anharmonic oscillation,
 similar to that found in the experiments (i.e., with much more
 rapid changes in the blue-shifting regions than in the intermedi-
 ate region). The change in R also gives a better agreement with
 the measured sensitivity to an increase in layer thickness d ;
 compare Figures 5 and 6c.

The main argument behind artificially increasing R is that
 the diameter of the nanodisks or nanoholes are not much smaller
 than the thickness of the dielectric layers in reality. This means
 that the point dipole approximation, which is the basis behind
 eq 2, underestimates the field strength close to the particle
 surface. Further, the $1/2d$ far-field component of A , which
 dominates for large d values, can be expected to underestimate
 the directionality of the reflected field (a disk would rather act
 as a “nanomirror” for $d \approx 2a \leq \lambda$). These effects taken together
 lead to a too weak particle-image coupling, which is partly
 compensated for by artificially increasing R .

In order to further understand the spectral behavior, we keep only the radiative $1/2d$ term in A and assume that the particle polarizability α is approximately a Lorentzian $\alpha(\omega) = \alpha_0\omega_0/(\omega_0 - \omega - i\gamma)$, where α_0 is the DC polarizability, which is proportional to the particle volume, ω_0 is the eigenfrequency, and γ is a damping constant. By substituting this expression into eq 3, one finds that the resonance position is given by $\omega/\omega_0 = 1 + (\alpha_0k^2R)/2d \cos(k2d)$, which shows that the LSPR position oscillates as a function of d with a periodicity close to $\lambda_{\max}/2n \approx 600/2 \times 1.5^2 = 200$ nm, where n is the refractive index of the dielectric layer. Similarly, the effective line-width is $\Gamma = \gamma - (\alpha_0k^2R\omega_0)/2d \sin(k2d)$, which exhibits maxima for $d \approx (2m - 1/2)\lambda_{\max}/4n$, $m = 0, 1, 2, \dots$. These distances correspond to the regions of maximum blue-shift, which explains the broadening found in Figures 3c and 4c.

4. Summary

We have studied the optical properties of short range ordered nanoholes and nanodisks as a function of overlayer thickness. By adsorbing 22-tricosenoic acid Langmuir–Blodgett films upon the nanostructures, we could follow the spectral variation up to a layer thickness of $d \approx 350$ nm. The LSPR peak position λ_{\max} exhibits an oscillatory behavior with a period of $\sim \lambda_{\max}/2n \approx 190$ nm, where n is the refractive index of the adsorbate. The spectral shifts versus coating thickness curves agree qualitatively with simulations based on a nanodisk polarizability renormalized through the image–dipole coupling model. However, the dipole model underestimates the coupling between the LSP and the interface, indicating the importance of finite size effects. The magnitude of the plasmon renormalization for large layer thicknesses, up to $\delta\lambda_{\max}/\delta d \approx 2$ at $d \approx 290$ nm, suggests that these thick dielectric layers could be utilized in designing new types of bio/chemo sensors. By embedding the metal nanostructure in, for example, a layer of SiO_2 of optimal thickness ($d \approx 100$ or 290 nm), it should be possible to monitor molecular adsorption on the glass surface through the change in the LSPR spectrum. This construct would have at least three advantages in comparison to the classical, metal surface based, LSPR sensor. First, the sample would be easier to reuse, because it is easier to clean a glass surface than the metal (for example using a Piranha solution). Second, the glass layer would protect the metal nanostructures from various aging problems, including oxidation and solvent-induced nanoparticle reshaping effects.³⁴ Third, there would be no chemical contrast between different regions of the sensor surface, which would greatly facilitate the functionalization steps necessary for bio/chemo sensing experiments.

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